

# Metal and alloy composites with carbon nanomaterials obtained by electrodeposition from deep eutectic solvents

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# Why these types of ionic liquids?

 $\Box$  A novel ionic liquid medium with interesting perspectives in metals and alloys electrodeposition has been recently developed, based on choline chloride (2-hidroxy-ethyl-trimethyl ammonium chloride, vitamin  $B_4$ , ChCl) eutectic mixtures with different hydrogen bond donor compounds, such as amides, glycols or carboxylic acids. These ionic liquids analogues proved to be cheap and easy to make, but they are environmentally friendly, too.

 $\Box$  Detailed investigations are still required to implement these novel electrodeposition procedures, in order to optimize operation parameters and deposits characteristics. Additional information in this field may significantly contribute to the extension of the practical applications of these systems.

 $\Box$  Metal and alloy composites with multiwall carbon nanotubes (**MWCNTs**) or graphene/graphene oxide (**GO**) related materials as coatings represent significant engineering materials suitable for a large range of industrial applications, due to their particular structures. The electrochemical route is of great interest for large scale manufacturing allowing the obtaining of various functional and protective composite coatings with lower cost and easier control of the process.

□ Therefore, some experimental results regarding the electrodeposition of **Co-MWCNTs**, **Ag-MWCNTs**, **Sn-rGO** and **Sn-Ni-rGO alloy** composites involving ChCl based electrolytes are presented. These media also facilitate the formation of stable and homogeneous dispersions of multiwall CNTs or graphene oxide nanosheets, thus overcoming the agglomeration problems.

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- Y. Zhang, X. Lu, X. Feng, Y. Shi, X. Ji, Properties and applications of choline-based deep eutectic solvents, Progress in Chemistry 26 (6) (2013) 881-892.
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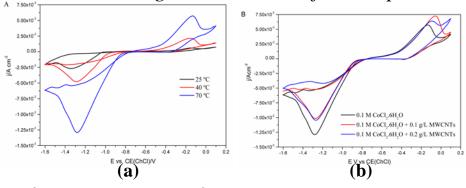
A 16-a ediție a Seminarului Național de Nanoștiință și Nanotehnologie București, Biblioteca Academiei Române, 6 iunie 2017

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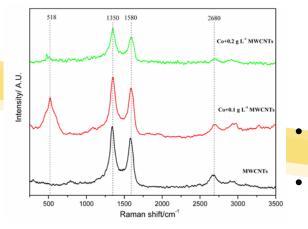
## 1. Electrochemical deposition of Co-MWCNTs composites

<u>Electrolyte:</u> choline chloride – urea (1:2) eutectic mixture +1M CoCl<sub>2</sub> + (0.1-0.5)g/L multi-walled carbon nanotubes (MWCNTs). Good dispersion of CNTs (6.9 nm diameter, 5  $\mu$ m length) by strong ultrasound stirring.

Establishing the conditions for composite electrodeposition



Characterization of Co-MWCNT coatings



**Fig. 2** Raman spectra for the **MWCNTs** (suspension in ChCl-urea) and **Co-MWCNTs** coatings prepared from ChCl-urea+1M CoCl<sub>2</sub>+0.1g/L MWCNTs (i =5mA/cm<sup>2</sup>, 15 min); ChCl-urea+1M CoCl<sub>2</sub>+0.2g/L MWCNTs (i =5mA/cm<sup>2</sup>, 15 min).

(b)

Pure MWCNTs dissolved in ChCl-urea showed Raman bands at 1380, 1580 and 2666 cm<sup>-1</sup> which are characteristic to the MWCNTs.

peak, but increased the anodic peak.

- Both cobalt containing samples gave also signals at Raman shifts assignable to MWCNTs, suggesting the successful formation of Co-MWCNTs composites.
- N.M. Pereira, O. Brincoveanu, A.G. Pantazi, C.M. Pereira, J.P. Araújo, A. F. Silva, M. Enachescu, L. Anicai, Electrodeposition of Co and Co composites with carbon nanotubes using choline chloride-based ionic liquids, Surf. Coat. Technol. DOI: 10.1016/j.surfcoat.2017.06.002
- M. Li, Z. Wang, R.G. Reddy, Cobalt electrodeposition using urea and choline chloride, Electrochim. Acta 123 (2014)325-331
- A. Cojocaru, M.L. Mares, P. Prioteasa, L. Anicai, T. Visan, Study of electrode processes and deposition of cobalt thin films from ionic liquid analogues based on choline chloride, J. Solid State Electrochem. 19 (4) (2015) 1001-1014.

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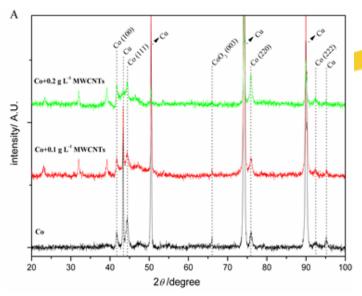
Fig. 1 CVs of Co-MWCNTs deposition on Cu substrate

from ChCl-urea +1M CoCl<sub>2</sub> +(0.1-0.2)g/L MWCNTs, 5mV/s:

(a) at different temperatures;

different concentration of MWCNTs at 70 °C.

The addition of MWCNTs has diminished the cathodic

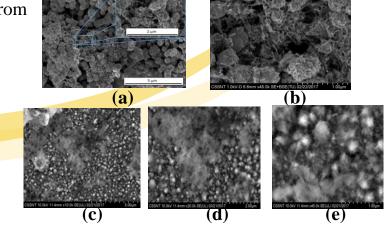


**Fig. 3** X-ray diffractograms for **Co** and **Co-MWCNTs** composite coatings prepared (i =5mA/cm<sup>2</sup>, 15 min) from: ChCl-urea+1M CoCl<sub>2</sub>; ChCl-urea+1M CoCl<sub>2</sub>+0.1g/L MWCNTs; ChCl-urea+1M CoCl<sub>2</sub>+0.2g/L MWCNTs.

- Literature shows that XRD patterns of MWCNTs evidence the strongest diffraction peak at the angle (2θ) of ≈25.5°, indexed as the (002) reflection of the <u>hexagonal graphite structure</u>. Existing differences for our Co-MWCNTs might be associated to deposit's internal stress and strain effects.
- Intensity of the peak assigned to Co(111) plane diminished and of Co(220) peak increased. This suggests that the presence of nanotubes promoted a preferential orientation of Co crystallites on (220).
- The inclusion of MWCNTs on the metal matrix modifies the grain dimensions and orientation.

Fig. 4 SEM images for the Co+MWCNTs prepared from ChCl-urea+1M CoCl<sub>2</sub>+0.2 g/L MWCNTs:
(a,b) 6.6 mA/cm<sup>2</sup>, 30 min., approx. 2.5 μm;
(c-e) 11 mA/cm<sup>2</sup>, 30 min., approx. 3.7 μm.

- Co-MWCNTs films show different morphology from Co films, mainly consisting in relatively spherical particles connected by MWCNTs (incorporated nanotubes).
- They appear to be less compact as compared to pure cobalt.



S. Costovici, A.-C. Manea, T. Visan, L. Anicai, Investigation of Ni-Mo and Co-Mo alloys electrodeposition involving choline chloride based ionic liquids, Electrochim. Acta, 207 (2016) 97-111.

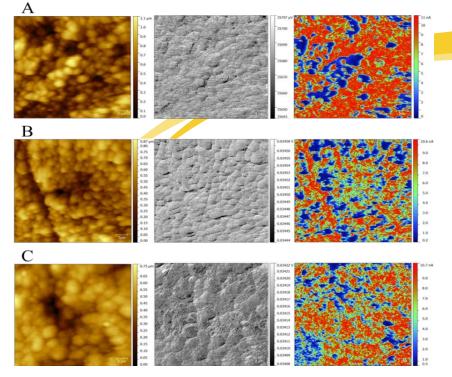


Fig. 5 AFM topographic images  $(a_1, b_1, c_1)$ , friction images  $(a_2, b_2, c_2)$ , and electrical conductivity maps  $(a_3, b_3, c_3, bias$ voltage – 1V) (20µm x 20 µm) for Co and Co-MWCNTs coatings prepared from ChCl-urea+1M CoCl<sub>2</sub> + (0.1-0.2)g/L MWCNTs (5mA/cm<sup>2</sup>, 15 min): (A) without MWCNTs; (B) +0.1g/L MWCNTs; (C) +0.2g/L MWCNTs. Surface of microarea: 20 µm x 20 µm

- The topography images show a surface formed by globular particles with diameters smaller than 600 nm. With such large height variations it is not possible to discern the nanotubes. Increasing the content of MWCNTs, the grains size diminished. Also, both average roughness ( $R_a$ ) and the root mean square (RMS) roughness diminished as the amount of nanotubes in electrolyte increased.
- Friction force maps roughly show that the local friction coefficient varies from high friction regions (**bright**) to low friction regions (**dark**). The absence of contrasting regions (Fig. 5(a<sub>2</sub>,b<sub>2</sub>,c<sub>2</sub>) may suggest that the friction force varies slightly along the surface.
- Electrical conductivity varies from high conductivity zones (**red**) to low conductivity or high resistive zones (**blue**). The composites have slightly lower conductivity when compared with pure cobalt film. The presence of amorphous carbon in the MWCNTs and their random spatial distribution in addition to the presence of cobalt oxide in the deposited layer might explain this decrease.

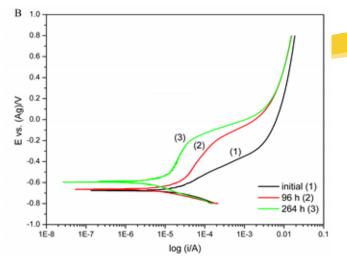


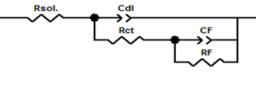
Table 1 Values of corrosion parameters from Tafel curves

		Pure Co		Co-MWCNT	
	Immersion time	E <sub>corr</sub> vs. Ag/V	i <sub>corr</sub> /μA cm <sup>-2</sup>	E <sub>corr</sub> vs. Ag/V	i <sub>corr</sub> /μA cm <sup>-2</sup>
I	Initial	-0.690	7.23	-0.677	7.17
	96 h	-0.642	7.07	-0.673	6.08
	264 h	-0.587	3.23	-0.604	4.93

**Fig. 6** Polarization curves (3mV/s) in 0.5M NaCl for **Co-MWCNTs** (prepared from ChCl-urea+1M CoCl<sub>2</sub>+0.2g/L MWCNTs (i=6.6mA/cm<sup>2</sup>, 30 min).

 $i_{corr}$  diminished as the immersion time increased which suggest the formation of a passive layer that improves the corrosion performance. Impedance plots evidence the presence of a single capacitive loop (Nyquist) and only one time constant in the middle frequency region (Bode).

Fig. 7 Equivalent-circuit proposed for Co-MWCNTs corrosion behavior in 0.5M NaCl aggressive solution.



**Table 2** Values of  $\mathbf{R}_{ct}$  (charge-transfer resistance for<br/>electrolyte/passive film interface) and  $\mathbf{R}_{F}$  (charge-<br/>transfer resistance for passive film / coating<br/>interface).

	,				
	Immersion time	Pure Co		Co-MWCNT	
		$R_{ct} (\Omega \ cm^2)$	$R_F(\Omega \text{ cm}^2)$	$R_{ct} (\Omega cm^2)$	$R_F(\Omega \ cm^2)$
	Initial	458	8753	595	4200
	24 h	10777	12536	224	4633
	48 h	4235	5603	2403	5000
	96 h	2454	1000	3650	10950
	168 h	329	8840	4236	11500

• N.M. Pereira, O. Brincoveanu, A.G. Pantazi, C.M. Pereira, J.P. Araújo, A. F. Silva, M. Enachescu, L. Anicai, Electrodeposition of Co and Co composites with carbon nanotubes using choline chloride-based ionic liquids, Surf. Coat. Technol. DOI: 10.1016/j.surfcoat.2017.06.002

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## 2. Deposition of Ag-MWCNTs composite coating

Establishing the conditions for composite electrodeposition from choline chloride-triethanolamine (IL-TEA) eutectic mixture

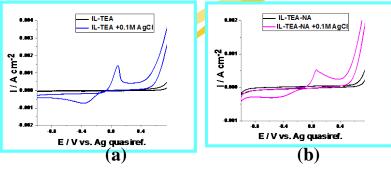


Fig. 8 Cyclic voltammograms of Ag<sup>+</sup>/Ag couple in:
(a) IL-TEA + 0.1M AgCl
(b) IL-TEA + 0.1M AgCl + NA;
GC working electrode, at 20mV/s, 70°C.

 Table 3 Electrolyte compositions and operating parameters

	Electrolysis condition	Coating characteristics		
Ag-TEA	0.1M AgCl dissolved in IL-TEA	1V, 70°C,	Adherent, non-glossy grey-yellow, 0.3µm,	
	(IL-TEA=choline	15-30 min.,	92.71-92,97 wt % Ag (EDX analysis)	
	chloride:triethanolamine (1:1)	300rpm magnetic stirring		
Ag-MWCNTs	0.1M AgCl +2g/L MWCNTs	$3,5-4V, 2 \text{ mA/cm}^2,$	Adherent, non-glossy (matte) grey,	
	suspensions in IL-TEA-NA	70°C, 15-30 min.	0.2-0.3μm,	
	(NA = 0.1M nicotinic acid)	with/whitout magnetic stirring	93.40-94,29wt % Ag (EDX analysis)	
Ag-MWCNTs	0.1M AgCl +2 g/L MWCNTs	Electrodeposition in ultrasonic bath,	Adherent, non-glossy (matte) grey,	
	suspensions in IL-TEA-NA	8.5 V, 2 mA/cm <sup>2</sup> , 70°C, 15-30 min	0.35-0.76 μm	
			MWCNTs evidenced by SEM analysis	
Ag-MWCNTs	0.1M AgCl +2 g/L MWCNTs	Electrodeposition in ultrasonic bath,	Adherent, non-glossy (matte) grey	
	suspensions in IL-TEA-NA	8.5 V, 4.8 mA/cm <sup>2</sup> ,	0.25-0.35 μm	
		70°C, 15-30 min	MWCNTs evidenced by SEM analysis	

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P. Sebastián, E. Vallés, E. Gómez, First stages of silver electrodeposition in a deep eutectic solvent. Comparative behavior in aqueous medium, Electrochim. Acta. 112 (2013) 149–158.

#### Characterization of Ag-MWCNTs coatings

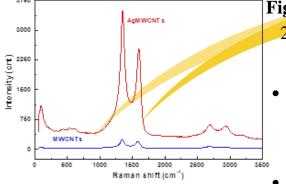
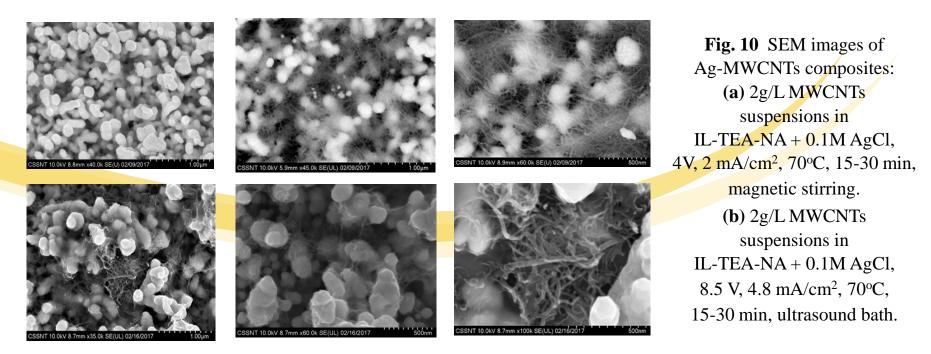


Fig. 9 Raman spectrum of Ag-MWCNTs coating (with green laser) prepared from 2g/L MWCNT s suspensions in IL-TEA-NA + 0.1M AgCl; 8.5 V, 4.8 mA/cm<sup>2</sup>, 70°C,15-30 min, ultrasound bath

- The presence of Raman bands characteristic to graphene: **D** band (1360 cm<sup>-1</sup>) and **G band** (1593 cm<sup>-1</sup>). The intensity of Raman bands increased with introduction of MWCNTs in coating, indicating a promising application for SERS measurements.
- The coatings morphology (SEM images) is in good agreement with the literature, but using aqueous electrolytes or other ionic liquids.



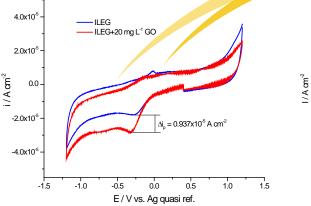
• G. Dunieskys, M.E.H. Larrude, M. da Costa, F. Freire Jr., Synthesis and characterization of silver nanoparticle-multiwalled carbon nanotube composites, J. Nanomaterials, 2014, Article ID 654068, 7 pages

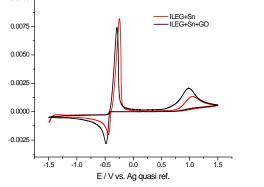
## 3. Deposition of Sn-rGO composite

Dispersion stability of graphene oxide nanosheets was excellent in the **choline chloride-ethylene glycol (ILEG**) eutectic solvent

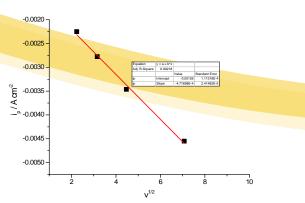
Cyclic voltammetry (CV) of  $Sn^{2+}/Sn$  couple on GC electrode

0.0100



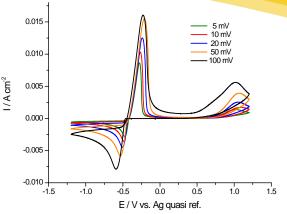


**Fig. 11** CVs on GC for ILEG and ILEG+20mg/L GO, 25°C, 20 mV/s



**Fig. 14** Dependence of peak current density on scan rate for ILEG+50mM SnCl<sub>2</sub>+20mg/L GO **Fig. 12** Comparison of **Sn** and **Sn+rGO** deposition on GC from ILEG+50mM SnCl<sub>2</sub> and ILEG+50mM SnCl<sub>2</sub>+20mg/L GO, 25°C, 20 mV/s

The experimental results show a diffusion-controlled cathodic process of **Sn-rGO** deposition from choline chloride – ethylene glycol +  $SnCl_2$  + graphene oxide bath, at room temperature.

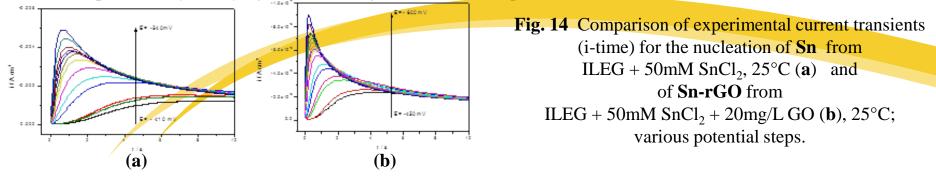


**Fig. 13 Sn+rGO** deposition on GC, from ILEG+50mM SnCl<sub>2</sub>+20mg/L GO, 25°C, different scan rates.

**Table 4** Values of  $i_p$  from CVs with variousscan rates (v) for ILEG+50mM SnCl<sub>2</sub> andILEG+50mM SnCl<sub>2</sub>+20mg/L GO, 25°C

	i <sub>p</sub> (mA/cm <sup>2</sup> )		
v (mV/s)	ILEG +50 mM	ILEG+50mM	
	SnCl <sub>2</sub>	SnCl <sub>2</sub> +20mg/LGO	
5	2.25	2.83	
10	2.77	3.61	
20	3.46	4.47	
50	4.55	5.82	
100	4.85	7.89	

#### Chronoamperometry study of nucleation for Sn+rGO deposits on GC electrode



We made a comparison of experimental data with theoretical curves  $(i/i_{max} - t/t_{max})$  corresponding to instantaneous nucleation (eq.1) or progressive nucleation (eq.2) [Scharifker]:

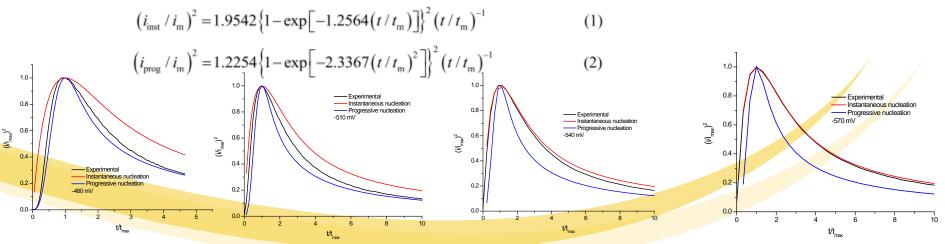
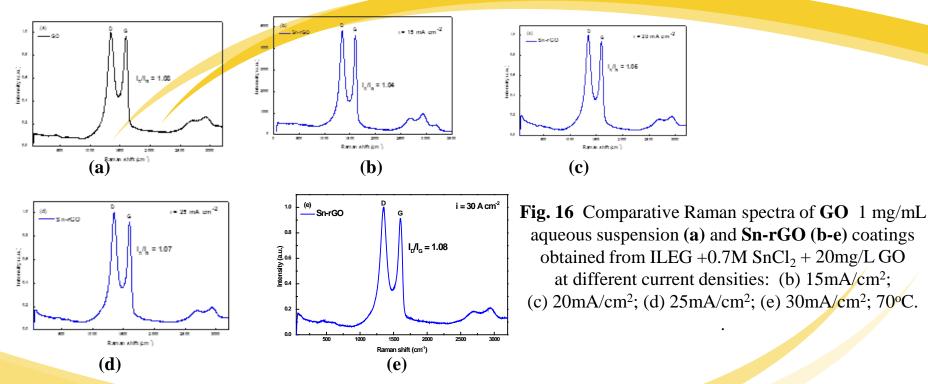


Fig. 15 Successive current transients for the nucleation of Sn-rGO showing gradual transition from progressive to instantaneous nucleation.

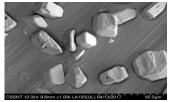
The time  $t_{max}$  for **Sn-rGO** is shorter (from 4.3s to 0.2s) when the cathodic polarization increases. The nucleation rate linearly increased [Ji] with the applied overpotential, having the slope higher than that for **Sn** coating.

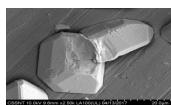
- B.R. Scharifker, G.J. Hills, Theoretical and experimental studies of multiple nucleation, Electrochim. Acta 28 (1983) 879-889
- C. Ji, G. Oskam, P. Searson, Electrochemical nucleation and growth of copper on Si(111), Surface Science 492 (2001) 115-124.

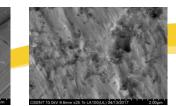
#### Characterization of Sn-rGO composite coatings



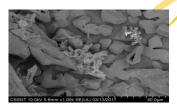
- Two peaks at about 1361cm<sup>-1</sup>(**D** band, k-point phonons of A1g symmetry) and 1584cm<sup>-1</sup>(**G** band, E2g phonon of Csp2 atoms) are observed. D/G intensity ratios in Sn-rGO coating are calculated to be 1.03-1.08.
- The increased D/G intensity ratio in Sn-rGO as compared to GO is due to a decrease in the average size of the sp<sup>2</sup> domains and an increase in the number of these domains, suggesting the reduction of GO to rGO in the electrodeposited composite [Wang, Feng].
- G. Wang, B. Wang, X. Wang, J. Park, S. Dou, H. Ahn, K. Kim, Sn/graphene nanocomposite with 3D architecture for enhanced reversible lithium storage in lithium ion batteries, J. Materials Chem. 19 (2009) 8378-8384.
- B. Feng, J. Xie, G. Cao, T. Zhu, X. Zhao, Facile synthesis of ultrafine CoSn nanocrystals anchored on graphene by one-pot route and the improved electrochemical Li-storage properties, New Journal of Chemistry 37 (2013) 474.

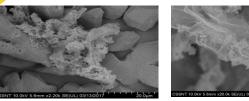




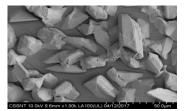


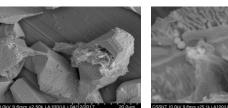
**Sn**, 15mA/cm<sup>2</sup>, 30 min.



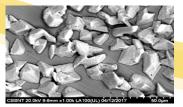


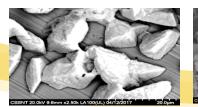
**Sn-rGO**, 15mA/cm<sup>2</sup>, 30 min.





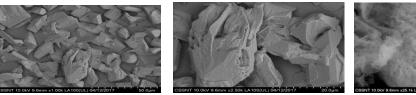
**Sn-rGO**, 20mA/cm<sup>2</sup>, 30 min.







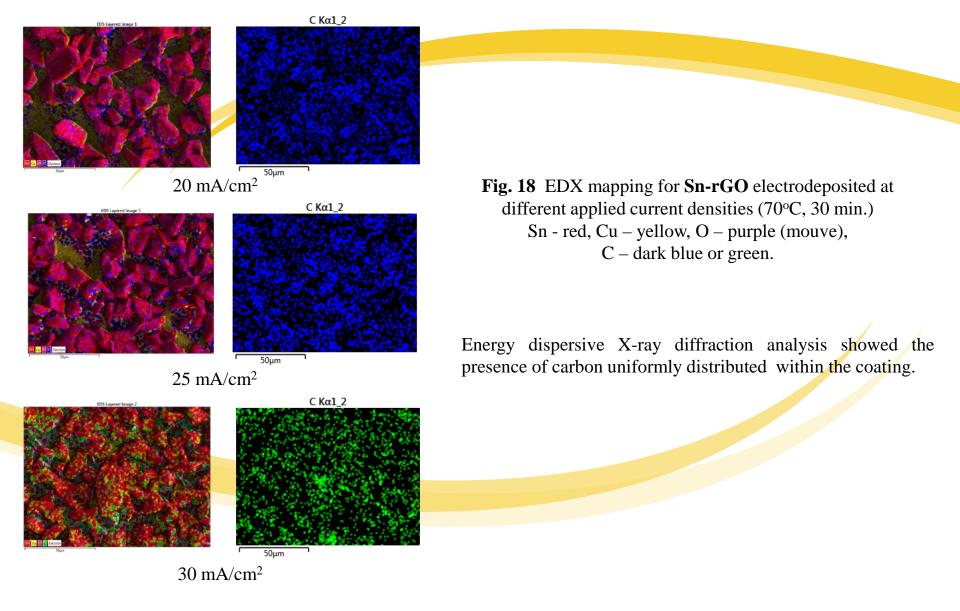
**Sn-rGO**, 25mA/cm<sup>2</sup>, 30 min.



**Sn-rGO**, 30mA/cm<sup>2</sup>, 30 min.

Fig. 17 Comparative SEM micrographs of Sn electrodeposited from ILEG +  $0.7M \text{ SnCl}_2$ and Sn-rGO electrodeposited from ILEG + $0.7M \text{ SnCl}_2 + 20mg/L \text{ GO}$ , different current densities; 70°C, magnetic stirring of bath.

The recorded SEM micrographs showed that the rGO incorporation within the deposit needs current density values higher than  $15 \text{ mA/cm}^2$ , in agreement with [Salome et al.].



• S. Salomé, N.M. Pereira, E.S. Ferreira, C.M. Pereira, A.F. Silva, Tin electrodeposition from choline chloride based solvent: Influence of the hydrogen bond donors, J. Electroanal. Chem. 703 (2013) 80–87.

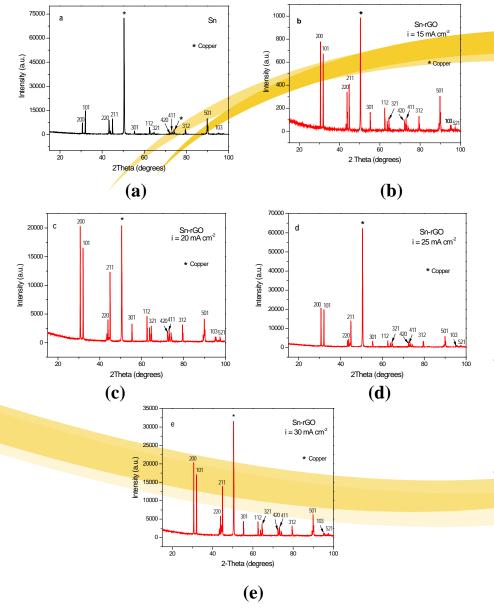
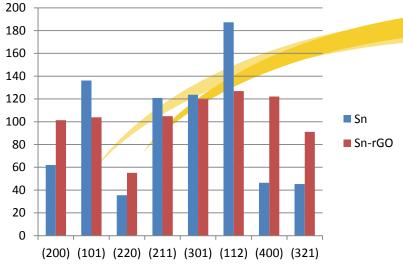
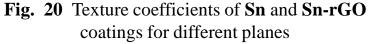


Fig. 19 Comparative X-ray diffractograms of Sn electrodeposited from ILEG + 0.7M SnCl<sub>2</sub> (a) and Sn-rGO electrodeposited from ILEG + 0.7M SnCl<sub>2</sub> + 20mg/L GO at different current densities: (a) 15mA/cm<sup>2</sup>; (b) 15mA/cm<sup>2</sup>; (c) 20mA/cm<sup>2</sup>; (d) 25 mA/cm<sup>2</sup>; (e) 30mA/cm<sup>2</sup>.

- XRD patterns of Sn and Sn-rGO coatings show clearly the diffraction peaks of crystalline Sn nanoparticles. They can be indexed to the tetragonal Sn phase (S.G.: I41/amd(141)), indicating the crystalline nature of the Sn-rGO coating.
- The peak at  $2\theta=23-28^{\circ}$  usually attributed to the graphene nanosheets is not prominent.

• *L.P. Bérubé, G. L'Espérance, A quantitative method of determining the degree of texture of zinc electrodeposits, J Electrochem Soc 136 (1989)* 2314-2315.





The preferred orientation of Sn and Sn-rGO was estimated from X-ray data according to the methodology developed by Berube and L'Esperance. The texture coefficient was calculated from the following equation:

$$Tc(hkl) = \frac{I(hkl)}{\sum I(hkl)} \times \frac{\sum Io(hkl)}{Io(hkl)} \times 100$$
(3)

where:  $I_{(hkl)}$  is the peak intensity of the electrodeposited Sn and  $I_0$  is the intensity of peak of standard Sn powder sample from JCPDS file card 004-0673.

# Table 5 Texture coefficient of Sn and Sn-rGO coatingsSamples20<br/>(degrees)Crystallite size,<br/>(nm)Tc<br/>(hkl)

Samples	20	h k l plane	Crystallite size,	Tc <sub>(hkl)</sub>	
Sumptos	(degrees)	n n i piune	(nm)	• C(hkl)	
Sn coating	30.63	200	77.9	62.01	
	32.01	101	78.5	136.14	
	43.87	220	84.8	35.51	
	44.90	211	85.4	120.93	
Sn-rGO coating	30.63	200	76.4	101.35	
$Ic = 20 mAcm^{-2}$	31.99	101	76.5	103.91	
	43.83	220	78.5	55.19	
	44.90	211	78.5	104.91	

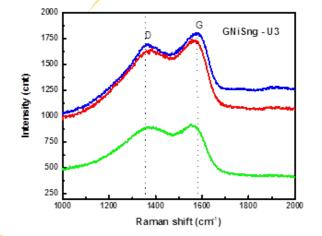
- The average sizes of cristallites have been estimated to be around 81 nm for Sn and 77 nm for Sn-rGO deposits using Debye-Sherrer' equation regardless the applied current density.
- A more pronounced orientation along the planes (101), (211), (301) and (112) has been noticed in the case of pure Sn, while the presence of rGO in the deposit diminished this tendency.

*L.P. Bérubé, G. L'Espérance, A quantitative method of determining the degree of texture of zinc electrodeposits, J. Electrochem. Soc. 136 (1989)* 2314-2315.

## 4. Electrochemical deposition of Sn-Ni + rGO composite alloys

Applications of Sn-Ni: as soldering alloys [Liu] and as electrodes in Li-ion batteries [Yang]

**Sn-Ni-rGO** composite coatings (1µm) were obtained applying current densities in the range of 30-80 mA/cm<sup>2</sup> for 30 min. under stationary conditions or ultrasound stirring, at temperatures between 30-50°C. <u>The electrolyte</u>: **choline chloride-ethylene glycol eutectic mixture** (denoted ILEG) + (0.1-2)M total concentrations of  $(SnCl_2+NiCl_2) + 1-2$  g/L GO.



- The Raman bands characteristic to graphene: **D** band (1360 cm<sup>-1</sup>) and **G band** (1593 cm<sup>-1</sup>) were noticed [Jiang].
- Adherent, uniform **Sn-Ni-rGO** composite coatings have been electrodeposited.
- According to the literature, the electrochemical reduction of **GO** to **rGO** takes place simultaneously with the alloy deposition.

**Fig. 21** Raman spectra of **Sn-Ni-rGO** composite alloys prepared under various hydrodynamic conditions (stationary or ultrasound stirring).

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- *H. Yang, L. Li, Tin–indium/graphene with enhanced initial coulombic efficiency and rate performance for lithium ion batteries, J. Alloys Compd. 584 (2014)* 76–80.
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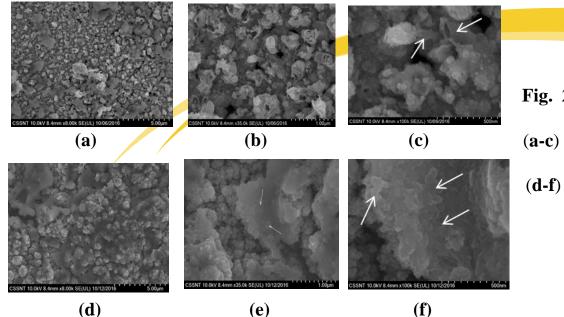


Fig. 22 SEM images of Sn-Ni-rGO composite alloys prepared at 50°C, from: (a-c) ILEG + 67mM SnCl<sub>2</sub> + 33mM NiCl<sub>2</sub> + 2g/L GO, 50 mA/cm<sup>2</sup>, 30 min.; (d-f) ILEG + 0.25M SnCl<sub>2</sub>+ 0.25M NiCl<sub>2</sub> + 2g/L GO, 30 min., ultrasound stirring.

- Adherent, uniform Sn-Ni-rGO composite coatings have been electrodeposited.
- The composite coatings contained about 67-70% Sn according to EDX analysis, with a very slight variation against the applied current density domain.
- The additional application of ultrasonic stirring of the electrolyte significantly increased the layer thickness of the deposited composite up to 10-12 μm at 30-40 mA/cm<sup>2</sup> for 30 min.
- Preliminary XRD studies showed that addition of graphene oxide in Sn-Ni alloy matrix determined a decrease in the grain sizes. The XRD patterns (not shown here) exhibited a diffraction peak at around 25°, however of a lower intensity as compared to other peaks.
- According to XRD patterns, the electrochemical deposited **Sn-Ni-rGO** composite coatings show a nanocristalline structure, with average sizes of the cristallites around 11-14.5 nm.

## **Concluding remarks**

- Co-MWCNTs, Ag-MWCNTs, Sn-rGO and Sn-Ni-rGO composite alloys have been successfully electrodeposited from choline chloride based ionic liquids analogues.
- Preliminary investigations (generally involving electrochemical techniques) allowed us to optimize the operating parameters for composites electrodeposition.
- The physical characterization of composite coatings using Raman spectroscopy, SEM (AFM) microscopy and XRD confirmed the incorporation of CNTs and graphene oxide nanosheets (GO), respectively. For composite alloys containing graphene oxide the electrochemical reduction of GO to rGO is supposed to take place during the cathodic process.
- Raman spectra exhibit specific bands for the corresponding MWCNTs or GO species. In addition, the recorded Raman spectra in the case of Ag-MWCNTs composites suggest their application for SERS analysis.
- The SEM microscopy evidenced the presence of the involved carbon nanomaterial incorporated within the metallic deposit.
- The recorded XRD patterns showed that the MWCNTs or rGO presence within the metallic matrix modifies the grain orientation and diminishes the grain sizes of the metallic species.

#### Acknowledgements

Part of this work was supported by the Romanian Ministry of Education, EXECUTIVE AGENCY FOR HIGHER EDUCATION, RESEARCH, DEVELOPMENT AND INNOVATION FUNDING, under NANOCOATIL project 7-082/2013 and NOVTINALBEST project 38/2016, both from M Era Net Program

# Thank you very much for your attention !